

## Preparation and Properties of Iron(II) Hydroxide-(Sodium Fluoride Tetrasilicic Mica) Intercalation Complexes

Kunio OHTSUKA,\* Mitsuru SUDA, and Mikiya ONO

Research & Development Center, Ceramics, Mitsubishi Mining & Cement Co., Ltd.,  
2270 Yokoze, Chichibu-gun, Saitama 368

(Received July 7, 1987)

Iron(II) hydroxide-(sodium fluoride tetrasilicic mica) intercalation complexes, which have a chlorite-like structure, were prepared by titrating iron(II) sulfate solutions in the presence of mica with sodium hydroxide under oxygen-free conditions. The formation of iron(II) hydroxide sheets between the silicate layers was evaluated by chemical analysis, XRD, DTA-TGA, and surface-area measurement. The presence of water is essential for the oxidizing of these complexes to occur at room temperature by air, because the  $O_2$  attack on the surface of these iron(II) hydroxide sheets is hindered by the silicate layers. Fine highly dispersed iron metal particles ( $\alpha$ -Fe) were formed on the external surface of the silicate by the  $H_2$  reduction of the complexes.

We previously reported that ultra-fine, highly dispersed metal (Ni, Co) particles can be prepared on the external surface of silicates by the reduction of metal (Ni, Co) hydroxide-layer silicate complexes.<sup>1,2)</sup> This method of preparation, providing isolated metal atoms on the carrier, leads to many applications for catalysts, magnetic materials, electronics, etc. Therefore, it is desired that this technique is applicable to the iron element as well as to the nickel and cobalt elements; the iron element is essential for those applications.

Besides, iron(II) hydroxide-layer silicate complexes, if possible, should be novel reducing agents which are easy to handle, for iron(II) hydroxide has a strong reducing power and this iron(II) hydroxide sheet would be sandwiched between the silicate layers.

Although many attempts have been made to prepare these iron hydroxide-layer silicate complexes,<sup>3–7)</sup> no clear evidence for the formation of these complexes has yet been obtained. These attempts to study clay minerals have been connected with the transformation of expansible layer silicate (2:1 layer silicate) to metal hydroxide-layer silicate, which has a chlorite-like structure. Chlorites have a 2:1 layer structure with a metal hydroxide interlayer, and iron(II, III) is considered to exist in both the hydroxide sheet and the 2:1 layer of chlorites.<sup>8–12)</sup>

In this work, the preparation and properties of iron hydroxide-layer silicate (sodium fluoride tetrasilicic mica) complexes were investigated. We have attempted to prepare these complexes with the iron(II) ion, not with the iron(III) ion, because we consider that there is an essential difficulty in forming an iron hydroxide interlayer with the iron(III) ion; the iron(III) ion in the metal hydroxide interlayer of chlorites must be introduced by the oxidation of the iron(II) ion or by coprecipitation as a minor constituent with a major constituent (Al, Mg etc.). One of the reasons why the preparation of iron hydroxide-layer silicate complexes has rarely been carried out with the iron(II) ion is evidently that iron(II)-ion solutions are easily oxidized by air. The

problems of what kind of cations can give these hydroxide interlayers and of the mechanism of these transformations will be explained in detail in our next paper.

### Experimental

**Starting Materials.** A 10 wt% sol of sodium fluoride tetrasilicic mica ( $NaMg_{2.5}Si_4O_{10}F_2$ ), obtained from Topy Ind., Co., under the name of Na-TSM, was used. Since this mica sol contains a little  $\alpha$ -cristobalite and magnesium fluoride richterite ( $Na_2Mg_6F_2(Si_4O_{11})_2$ ) as impurities, the sol was diluted with distilled water, and then most of the impurities were separated by centrifugation. Finally sodium fluoride tetrasilicic mica was dried in air at 100 °C.

The characteristics of this mica were described in detail in our previous work.<sup>1)</sup> Sodium fluoride tetrasilicic mica is an expansible 2:1 layer silicate.  $Na^+$  ions are located between the silicate layers to balance the negative layer charge of the  $-1$  derived from the vacancies in the silicate layers. These  $Na^+$  ions are exchangeable in water, the theoretical cation-exchange capacity being 254 meq/100 g·mica.

#### Preparation of Iron(II) Hydroxide-Mica Complexes.

Iron(II) hydroxide-(sodium fluoride tetrasilicic mica) complexes were prepared by the titration method: iron(II) sulfate solutions in the presence of mica were titrated with sodium hydroxide under oxygen-free conditions.

The iron(II) ion is thermodynamically unstable with respect to atmospheric oxidation and, especially in alkaline solutions, is rapidly oxidized by the oxygen in the air.<sup>13,14)</sup> Hence, great care was taken to prevent it from touching with  $O_2$  during the preparation. The preparation procedures were operated under a  $N_2$  gas atmosphere;  $N_2$  gas was passed through a disoxidation agent in advance. All the solutions were prepared using distilled water free of  $O_2$ ; dissolved oxygen was expelled by keeping  $N_2$  gas bubbling through the boiling water.

Figure 1 shows the process of preparing the complexes, while Fig. 2 shows the scheme of the preparation system. Oxygen-free water was added to the mica powder, and then this suspension was refluxed in a  $N_2$  flow to expel the oxygen absorbed on the surface of mica. As the iron(II) sulfate reagent has a few iron(III) ions, iron(II) sulfate solutions were aged for 1 day and then passed through glass filters by means of  $N_2$ -gas pressure to remove any iron(III) hydroxide. These iron(II) sulfate solutions, thus filtered and

transported to the reaction flask, were then mixed with the mica sol to give a 1 mol Fe/100 g-mica, where the concentration of Fe in these mixed suspensions was 0.1 M (1 M=1 mol dm<sup>-3</sup>). These mixed suspensions were stirred for 1 day to promote ion-exchange between the iron(II) ion and the sodium ion within the mica. Subsequently, in order for the hydrolysis between the silicate layers to proceed, these suspensions were titrated with 0.1 M sodium hydroxide solutions at the rate of 20 ml h<sup>-1</sup> to yield OH/Fe molar ratios of 0, 0.2, 0.4, 0.6, 0.8, and 1.0 respectively under

vigorous stirring. After these additions, stirring was continued for 2 days. The resulting products were washed in oxygen-free water to remove electrolytes, the supernatant being taken out by means of N<sub>2</sub>-gas pressure. Finally, these products were centrifuged, dried at 50 °C in a vacuum oven, and stored under nitrogen in a tightly closed desiccator.

**Techniques of Characterizing the Complexes.** Chemical analyses of the complexes were carried out by atomic-absorption analysis. The X-ray diffraction (XRD) patterns were obtained with a Rigaku diffractometer RU-200, using Cu K $\alpha$  radiation, from specimens basally oriented on glass slides. The surface areas were measured by the nitrogen-adsorption method at the temperature of liquid nitrogen with a Carlo Elba Sorptomatic, Series 1800. The samples were evacuated at 50 °C overnight prior to the measurements. Differential thermal analysis and thermogravimetric analysis (DTA-TGA) were made with a Rigaku Thermoanalyzer M8076 in a N<sub>2</sub> flow at a heating rate of 10 °C min<sup>-1</sup>. Transmission electron microscopy (TEM) was performed with a JEOL, Model JEM-1200EX. The thermal stability of the complexes in an inert gas and their reduction behavior with H<sub>2</sub> were investigated in a range of 100–700 °C at a heating rate of 10 °C min<sup>-1</sup>, because this mica is stable below 800 °C.

**Measurement of the Room-Temperature Oxidation of the Complexes in Air.** In our preliminary experiment, it was found that these complexes were not oxidized only by oxygen, but that the presence of water appeared to be essential for oxidizing to occur at room temperature by air. Therefore, the room-temperature oxidation of these complexes by air was investigated in an air flow with water vapor, the weight change during exposure to air being measured by means of the gravimetric McBain-Bakr balance technique. Samples of about 0.2 g were hung with a quartz spring and evacuated. Subsequently, N<sub>2</sub> gas with a saturated water-vapor pressure (at 30 °C) was passed over the sample (at 31 °C) at a flow rate of 100 ml min<sup>-1</sup>, the relative humidity being 94%. After a constant weight was achieved, the flow was then switched to air with a saturated water-

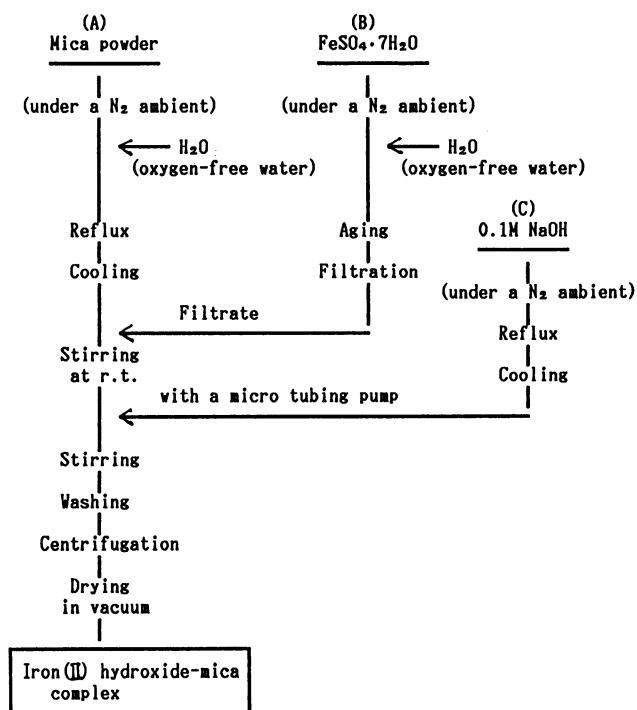


Fig. 1. Process of preparing iron(II) hydroxide-(sodium fluoride tetrasilicic mica).

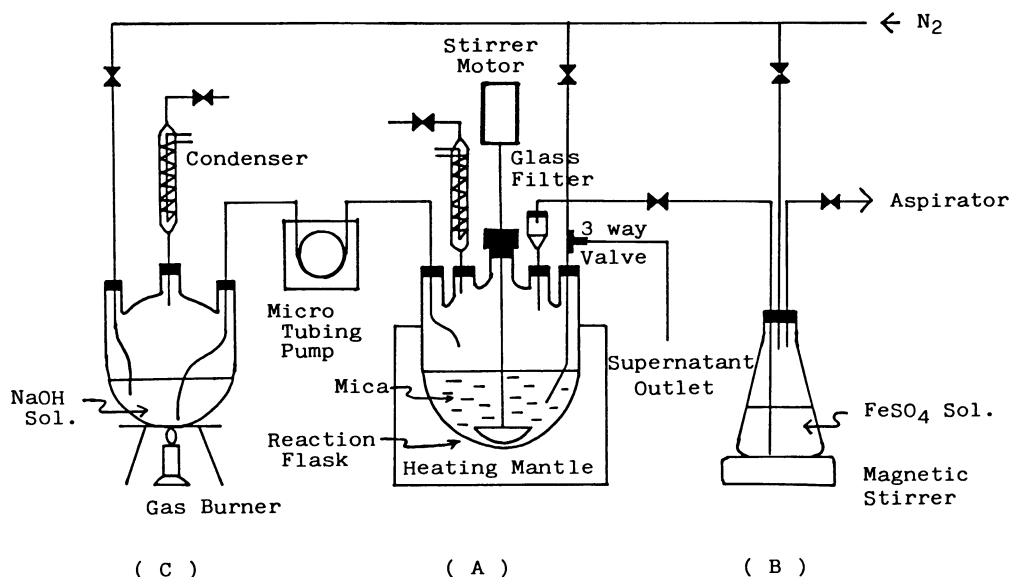


Fig. 2. Schematic of apparatus for preparing iron(II) hydroxide-(sodium fluoride tetrasilicic mica).

vapor pressure (at 30 °C) until the weight gain by oxidation was equilibrated.

### Results and Discussion

**Chemical Analyses.** The results of the chemical analyses of the mica and its intercalation complexes are listed in Table 1. The iron content increased with an increase in the OH/Fe ratio, and the extent of the ion-exchange ratio in the resulting products was almost constant (81–82%). At an OH/Fe ratio of 1.0, the iron content of the complexes was 2.4 mol per formula unit of the mica 82% ion-exchanged, corresponding to 2.9 mol per formula unit of the mica 100% ion-exchanged. The ideal composition of the iron (II) hydroxide-mica complexes, which have a chlorite-like structure, is  $[\text{Fe}_3(\text{OH})_5(\text{H}_2\text{O})](\text{Mg}_{2.5}\text{Si}_4\text{O}_{10}\text{F}_2)^{1,2}$ ; the maximum iron content of the complexes is 3 mol per formula unit of the mica 100% ion-exchanged. Therefore, at an OH/Fe ratio of 1.0, the iron(II) hydroxide interlayers in the complex must be well-developed.

### XRD Measurement and Specific Surface Area.

The X-ray powder diffraction patterns of these complexes are shown in Fig. 3. The basal spacings of these complexes expanded to about 14.3–14.7 Å from the 9.6 Å basal spacing of the mica (Table 2); these basal spacings were almost independent of the iron content. Moreover, with an increase in the OH/Fe

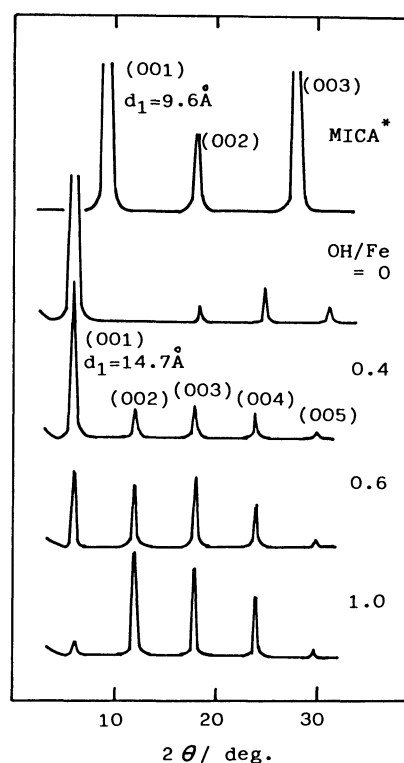


Fig. 3. X-Ray powder diffraction patterns of iron(II) hydroxide-(sodium fluoride tetrasilic mica). MICA\*: anhydrous form.

Table 1. Conditions of Preparation and Chemical Analyses of Sodium Fluoride Tetrasilic Mica and Iron(II) Hydroxide-(Sodium Fluoride Tetrasilic Mica)

No.	Condition of preparation	Chemical analysis/wt%			Molar ratio ( $\text{SiO}_2=4$ )			Ion exchange ratio/% <sup>b</sup>
	OH/Fe	$\text{SiO}_2$	Na	Fe	$\text{SiO}_2$	Na	Fe	
MICA <sup>a</sup>		57.34	4.36	—	4.00	0.80	—	—
1	0.0	53.51	0.96	3.29	4.00	0.19	0.26	81
2	0.2	50.43	0.89	8.10	4.00	0.18	0.69	82
3	0.4	47.38	0.82	12.40	4.00	0.18	1.13	82
4	0.6	44.55	0.77	15.98	4.00	0.18	1.54	82
5	0.8	41.94	0.71	19.72	4.00	0.18	2.02	82
6	1.0	39.36	0.67	21.95	4.00	0.18	2.40	82

a) Sodium fluoride tetrasilic mica ( $\text{NaMg}_{2.5}\text{Si}_4\text{O}_{10}\text{F}_2$ ). Interlayer charge is +1 per formula unit. Initial interlayer cations are  $\text{Na}^+$ ; part of these  $\text{Na}^+$  ions are displaced by  $\text{H}^+$  ions when the silicate sol is diluted in distilled water to eliminate impurities. b) Ion-exchange ratio is calculated by means of:  $(1 - \text{Na molar ratio}) \times 100$ .

Table 2. Basal Spacings (Å) of Iron(II) Hydroxide-(Sodium Fluoride Tetrasilic Mica) After Heat Treatment

No.	OH/Fe	Basal spacing/Å			
		50 °C <sup>a</sup> in a vacuum	200 °C <sup>b</sup> in Ar	300 °C <sup>b</sup> in Ar	500 °C <sup>b</sup> in Ar
Initial mica		9.6	9.6	9.6	9.6
1	0.0	14.3	9.7	9.6	9.6
2	0.2	14.7	9.7	9.6	9.6
3	0.4	14.7	ML <sup>c</sup>	ML	9.6
4	0.6	14.7	14.6	ML	9.6
5	0.8	14.7	14.7	14.6	9.6
6	1.0	14.7	14.7	14.6	9.6

a) Drying in a vacuum for 24 h. b) Heat-treatment in Ar for 2 h. c) ML=mixed layer (mixtures of 14.7 Å and 9.6 Å layers).

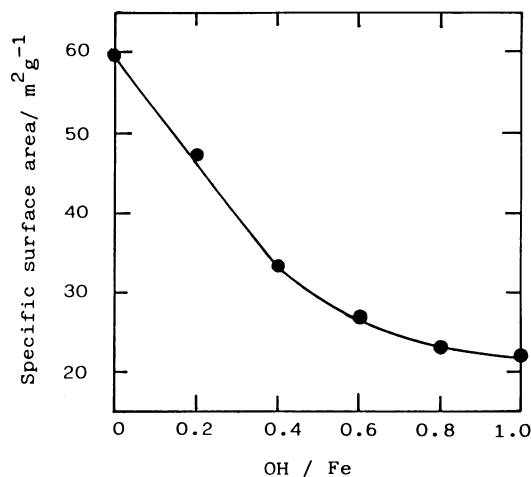


Fig. 4. Specific surface areas of iron(II) hydroxide-(sodium fluoride tetrasilicic mica).

ratio, the (001) first-order reflection decreased in intensity; on the contrary, the (002)–(004) higher-order reflections increased in intensities. These changes in the relative intensities were the same as those observed in the formation reactions of nickel(II) or cobalt(II) hydroxide–mica complexes, indicating that an iron(II) hydroxide-forming reaction took place between the silicate layers. The XRD patterns of the complexes with OH/Fe ratios of 0.8 and 1.0 showed the essential X-ray characteristics of chlorites. Moreover, XRD analysis revealed the absence of any iron oxide in these heat-treated samples.

The specific surface areas of the complexes are shown in Fig. 4. With an increase in the OH/Fe ratio, the specific surface areas decreased because the development of iron(II) hydroxide between the silicate layers begins at the cation-exchange sites, which are apart from each other, and finally the internal spaces are completely stacked with this iron(II) hydroxide interlayer. These changes are the same as those observed in nickel(II) or cobalt(II) hydroxide–mica complexes.

**Thermal Analysis.** The DTA-TGA curves for the complexes are shown in Fig. 5. Complexes with an OH/Fe ratio of 0.0 showed a single endothermic peak at about 170 °C. This dehydration is due to the hydration water associated with the iron(II) ions introduced by ion-exchange between the silicate layers. The composition of these iron(II) ions appears to be  $[\text{Fe}(\text{OH})_6(\text{H}_2\text{O})_{6-\delta}]^{2-\delta}$ , because the hydration water of the cation in expandable layer silicates is much more dissociated than in liquid water.<sup>15)</sup> However, as the OH/Fe ratio increased, another peak appeared at about 375 °C, while the peak at about 170 °C was diminished. These findings indicate that iron(II) hydroxide  $[\text{Fe}_x(\text{OH})_y(\text{H}_2\text{O})_m]^{2x-y}$  developed between the silicate layers with the progressive

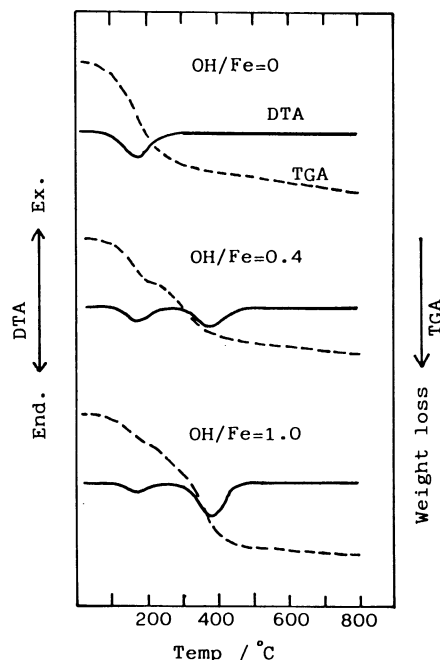


Fig. 5. DTA-TGA curves, obtained using a nitrogen atmosphere, for iron(II) hydroxide-(sodium fluoride tetrasilicic mica).

hydrolysis and olation of iron(II) cations by the addition of alkali. Therefore, the second peak corresponds to the loss of the hydroxyl group.

Table 2 shows the changes in the basal spacings after heat treatment in an Ar flow. Complexes with OH/Fe ratios of 0.8 and 1.0, which have a well developed iron(II) hydroxide interlayer, did not collapse even upon heat treatment at 300 °C. As iron(II) hydroxide decomposes to FeO at about 200 °C,<sup>16)</sup> well-developed iron(II) hydroxide between the silicate layers has a higher thermal stability than iron(II) hydroxide in the bulk state and is strongly fixed to the silicate layers.

**Reduction Behavior with H<sub>2</sub>.** The complexes were heated in a H<sub>2</sub> flow at 400–700 °C for 2 h. The XRD patterns of the reduced complexes with an OH/Fe ratio of 1.0 are shown in Fig. 6. These complexes collapsed to a basal spacing of 9.6 Å, and metallic iron ( $\alpha$ -Fe) was detected above 500 °C. As the reducing temperature rose, the reflection intensity of iron increased and the (001) mica basal reflection peak became sharper: reduced iron atoms gradually migrated from the interlayer of the silicate to its external surface.

A TEM photograph of the H<sub>2</sub>-reduced complexes with an OH/Fe ratio of 1.0 at 700 °C is shown in Fig. 7. Metallic iron particles with a clear cubic crystal habit are distributed on the external surface of the mica. The grain diameter of these metallic iron particles increased with an increase in the reducing temperature.

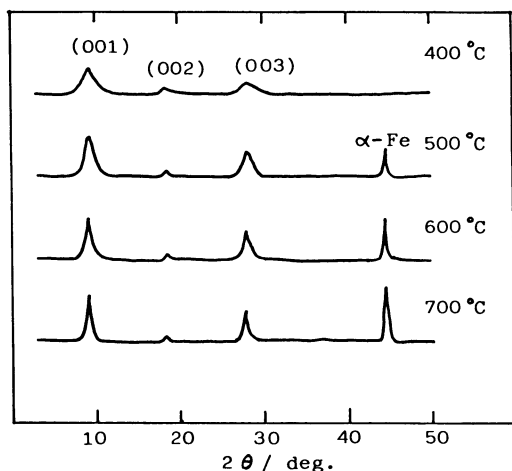


Fig. 6. X-Ray powder diffraction patterns of  $H_2$  reduced iron(II) hydroxide-(sodium fluoride tetrasilicic mica) with a OH/Fe ratio of 1.0 at various temperatures.

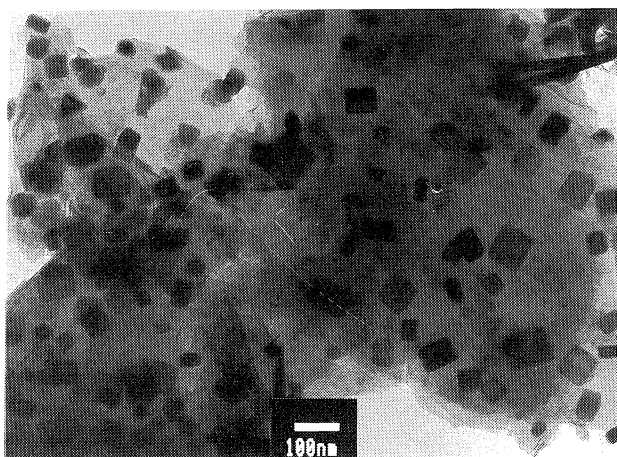


Fig. 7. Transmission electron micrograph of  $H_2$  reduced iron(II) hydroxide-(sodium fluoride tetrasilicic mica) with a OH/Fe ratio of 1.0 at 700 °C.

**Room-Temperature Oxidation of the Complexes in Air.** Figure 8 shows the room-temperature oxidation of the complexes with time in air with water vapor ( $P_{H_2O}=4.24$  kPa); temp=31 °C, R.H.=94%, air flow rate=100 ml min<sup>-1</sup>. The weight-gain values are plotted against the time in the logarithmic scale. Oxidation occurred gradually, and weight change by oxidation was large in the complexes with high OH/Fe ratios. The final weight-gain of the complexes with an OH/Fe ratio 1.0 was about 5 wt%. With the progress of oxidation, the color of the complexes changed from blue-green to yellow-orange and the chlorite-like structure was destroyed (Fig. 9). The oxidation rate was largely governed by the relative humidity and the reaction temperature, being controlled by the partial isomorphous substitution of

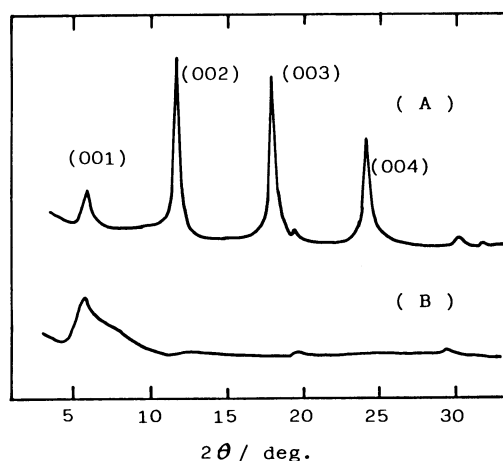


Fig. 9. X-Ray powder diffraction patterns of iron(II) hydroxide-(sodium fluoride tetrasilicic mica) after oxidation in air.

(A): original iron(II) hydroxide-(sodium fluoride tetrasilicic mica) complex with a OH/Fe ratio of 1.0; (B): oxidized complex in air with water vapor ( $P_{H_2O}=4.24$  kPa; R.H.=94%) at 31 °C for 2 d.

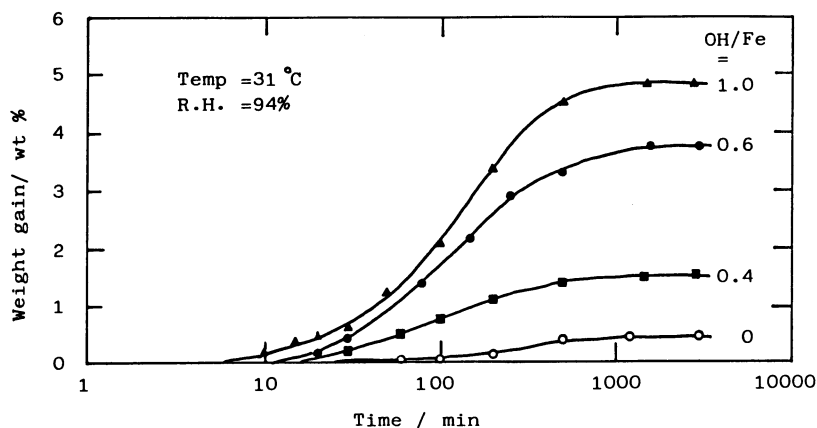


Fig. 8. Weight gain for iron(II) hydroxide-(sodium fluoride tetrasilicic mica) with time in air with water vapor ( $P_{H_2O}=4.24$  kPa; R.H.=94%) at 31 °C. R.H.= relative humidity.

- 1) K. Ohtsuka, M. Suda, M. Ono, M. Takahashi, M. Sato, and S. Ishio, *Bull. Chem. Soc. Jpn.*, **60**, 871 (1987).
- 2) K. Ohtsuka, J. Koga, M. Suda, M. Ono, and M. Takahashi, *Bull. Chem. Soc. Jpn.*, **60**, 2843 (1987).
- 3) D. D. Carstea, *Clays Clay Miner.*, **16**, 231 (1968).
- 4) D. D. Carstea, M. E. Harward, and E. G. Knox, *Soil Sci. Soc. Am. Proc.*, **34**, 517 (1970).
- 5) D. D. Carstea, M. E. Harward, and E. G. Knox, *Soil Sci. Soc. Am. Proc.*, **34**, 522 (1970).
- 6) S. Caillere and S. Henin, *Mineral. Mag.*, **28**, 612 (1949).
- 7) R. Herrera and M. Peech, *Soil Sci. Soc. Am. Proc.*, **34**, 740 (1970).
- 8) G. J. Ross and H. Kodama, *Clays Clay Miner.*, **24**, 183 (1976).
- 9) L. Makumbi and A. J. Herbillon, *Bull. Groupe Fr. Argiles*, **24**, 153 (1972).
- 10) G. J. Ross, *Nature (London)*, **225**, 133 (1975).
- 11) G. J. Ross and H. Kodama, *Clays Clay Miner.*, **22**, 205 (1974).
- 12) B. A. Goodman and D. C. Bain, "Proc. Int. Clay Conference, Oxford, 1978," ed by M. M. Mortland and V. C. Farmer, Elsevier, Amsterdam (1979), p. 65.
- 13) M. Pourbaix, "Atlas of Electrochemical Equilibria in Aqueous Solutions," Pergamon Press, Oxford (1966), p. 307.
- 14) D. Nicholls, "Comprehensive Inorganic Chemistry," ed by J. C. Bailar, H. J. Emeleus, Sir Ronald Nyholm, and A. F. Trotman-Dickenson, Pergamon Press, Oxford (1973), Vol. 3, p. 1005.
- 15) D. M. C. MacEvan and M. J. Wilson, "Crystal Structures of Clay Minerals and Their X-ray Identification," ed by G. W. Brindley and G. Brown, Mineralogical Society, London (1980), p. 219.
- 16) J. D. Bernal, D. R. Dasgupta, and A. L. Mackay, *Clay Min. Bull.*, **4**, 15 (1959).
- 17) W. Feitknecht, *Z. Elektrochem.*, **63**, 34 (1959).
- 18) M. Kiyama, *Bull. Chem. Soc. Jpn.*, **47**, 1646 (1974).
- 19) T. Sato, S. Okamoto, and K. Hashimoto, *Yogyo Kyokai Shi*, **94**, 1201 (1986).
- 20) O. K. Borggaard, H. B. Lindgreen, and S. Morup, *Clays Clay Miner.*, **30**, 353 (1982).
- 21) N. Kohyama, S. Simoda and T. Sudo, *Clays Clay Miner.*, **21**, 229 (1973).
- 22) C. C. Addison, W. E. Addison, G. H. Neal, and J. H. Sharp, *J. Chem. Soc., Pt. II*, **1962**, 1468.
- 23) G. W. Brindley and R. F. Youell, *Miner. Mag.*, **30**, 57 (1953).